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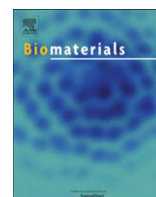


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Randomly oriented, upright SiO₂ coated nanorods for reduced adhesion of mammalian cells

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ARTICLE INFO

Article history:

Received 6 March 2009

Accepted 10 May 2009

Available online 10 June 2009

Keywords:

Nanorods

Cell adhesion

Cell survival

Anti-fouling

ABSTRACT

Cell interactions with nanostructures are of broad interest because of applications in controlling tissue response to biomedical implants. Here we show that dense and upright SiO₂ coated nanorods nearly eliminate cell adhesion in fibroblasts and endothelial cells. The lack of adhesion is not due to a decrease in matrix protein adsorption on the nanostructures, but rather an inability of cells to assemble focal adhesions. Using spatially patterned nanorods, we show that cells display a preference for flat regions of the surface. Our results support a model in which interfering with nanoscale spacing of ligated integrins results in reduced cell adhesion and subsequent cell death. We propose that dense monolayers of nanorods are a promising nanotechnology for preventing mammalian cell fouling of biomaterials.

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1. Introduction

Anchorage-dependent cells need to attach and spread on solid surfaces for normal function [1–3]. Modulating cell adhesion and survival by tailoring the surface is a promising strategy that has applications in tissue engineering and biomedical implants [4–7]. Cell adhesion to surfaces involves the adhesion of integrin receptors to their ligands, followed by subsequent nanoscale clustering of ligated integrins [8,9]. Therefore, recent approaches to modulate cell adhesion have focused on controlling the nanoscale adhesion and clustering of integrins [8–15]. These approaches include spatially patterning adhesive ligands [8–11], modulating the nanotopography of the substrate [12–15], surface modification with biocompatible polymers [16,17], and controlling cell attachment on patterned structures [18–20]. Applications for controlling cell behavior with nanostructured and nano-patterned materials range from improving integration of titanium implants with bone [21,22], to developing polymer scaffolds that better mimic the extracellular matrix [7,23], to anti-fouling materials for preventing cell adhesion to biomedical implants [24–26]. In particular, the design of effective surfaces that prevent mammalian cell adhesion has remained a fundamental challenge [27].

There are relatively few studies that have explored the use of nanostructures for eliminating mammalian cell adhesion and survival. Some evidence suggests that nanostructured materials

can be developed to reduce protein adsorption [28–30] and to potentially decrease cell adhesion [13–15,31]. In a previous paper, we have shown that endothelial cells and fibroblasts are unable to adhere and survive on zinc oxide (ZnO) nanorods compared to flat ZnO substrates [13]. The advantage of ZnO nanorods is that they can be grown with solution-based crystallization techniques at low-temperature. Thus, the nanorods can be coated on surfaces of irregular geometries, and temperature sensitive materials such as stents. However, it is unclear if the dramatic decrease in cell adhesion and survival observed on ZnO nanorods is reproducible with similar nanorods but of a different material. The chemical nature of the nanorod surface is clearly important given that it can potentially influence protein adsorption. In addition, ZnO has the potential for having long-term toxicity to cells due to leaching into solution [32–34].

Silicon dioxide (SiO₂) based nanowires and nanoneedles have received recent attention for modulating cell adhesion [15,35]. Previous studies have shown that stem cells can survive for long periods of time on surfaces sparsely coated with SiO₂ nanowires [35]. Conversely, on comparatively denser SiO₂ nanoneedles, cell adhesion is decreased, suggesting their potential for anti-fouling surfaces [15]. However, the decrease in cell adhesion on nanoneedles was not observed to be as dramatic [15] as previously reported with ZnO nanorods [13].

Therefore, in this paper, we explored if SiO₂ nanorods with similar morphologies as the previously used ZnO nanorods can result in a similar dramatic decrease in mammalian cell adhesion and survival. Our observations provide further evidence that densely packed upright nanorods can be used to develop surfaces resistive to mammalian cell adhesion.

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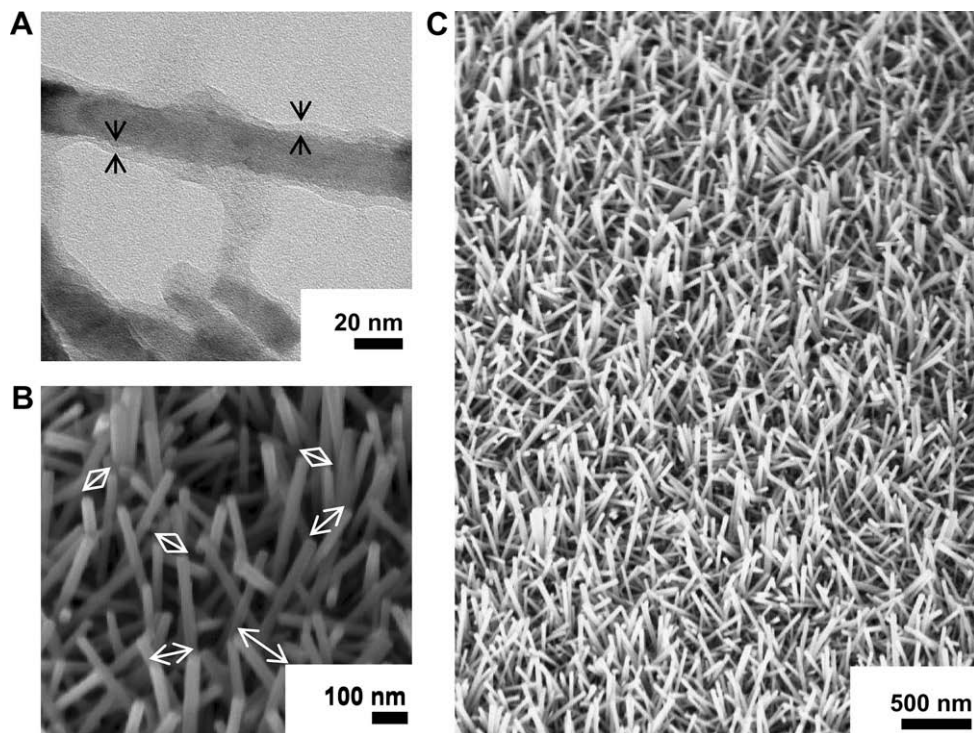


Fig. 1. The morphology of nanorods. (A) TEM image of SiO₂ deposited ZnO nanorods. Black arrows indicate SiO₂ thin film with a 50 Å thickness. ZnO nanorods are encapsulated by SiO₂. (B) Scanning electron microscopy (SEM) image of nanorods on glass. White arrows indicate the spacing between nanorods. The spacing between nanorods ranges from 80 to 100 nm. (C) SEM image of a monolayer of nanorods. Upright nanorods were covered on the underlying glass substrate uniformly.

2. Materials and methods

2.1. Fabrication of nanorods

ZnO nanorods were made by a solution-based hydrothermal growth method [36]. First, ZnO nanocrystal seed solutions were prepared by mixing 15 mM zinc acetate dihydrate (Sigma Aldrich, St. Louis, MO) with 30 mM of NaOH (Sigma Aldrich, St. Louis, MO) at 60 °C for 2 h. Next, ZnO nanocrystals were spin-coated onto the substrate and then post-baked on a hot plate at 200 °C for better adhesion. The substrate with these seeds was then suspended upside down in a Pyrex glass dish filled with an aqueous nutrient solution. The growth rate was approximately 1 μm per hour with 100 ml aqueous solution containing 20 mM zinc nitrate hexahydrate

and 20 mM hexamethylenetriamine (Sigma Aldrich, St. Louis, MO). To arrest the nanorod growth, the substrates were removed from solution, rinsed with de-ionized water and dried in air at room temperature. SiO₂ was deposited with a Unaxis 790 plasma enhanced chemical vapor deposition (PECVD) system at 50 °C using N₂O and 2% SiH₄ balanced by nitrogen as the precursors as reported before [37]. Patterned nanorods were fabricated by conventional photoresist (PR) lithography [36]. A glass slide was processed with negative PR (SU-8 2007, Microchem) so that a pattern with 50 μm circles was formed on the surface. The substrate was then post-baked at 110 °C for 30 min. The processed substrate was spin-coated with ZnO nanocrystals as seed materials and nanorods were grown on the substrate with an aqueous nutrient solution. The negative PR was removed by PG remover in a warm bath at 60 °C for 30 min. Patterned nanorods were also coated with SiO₂.

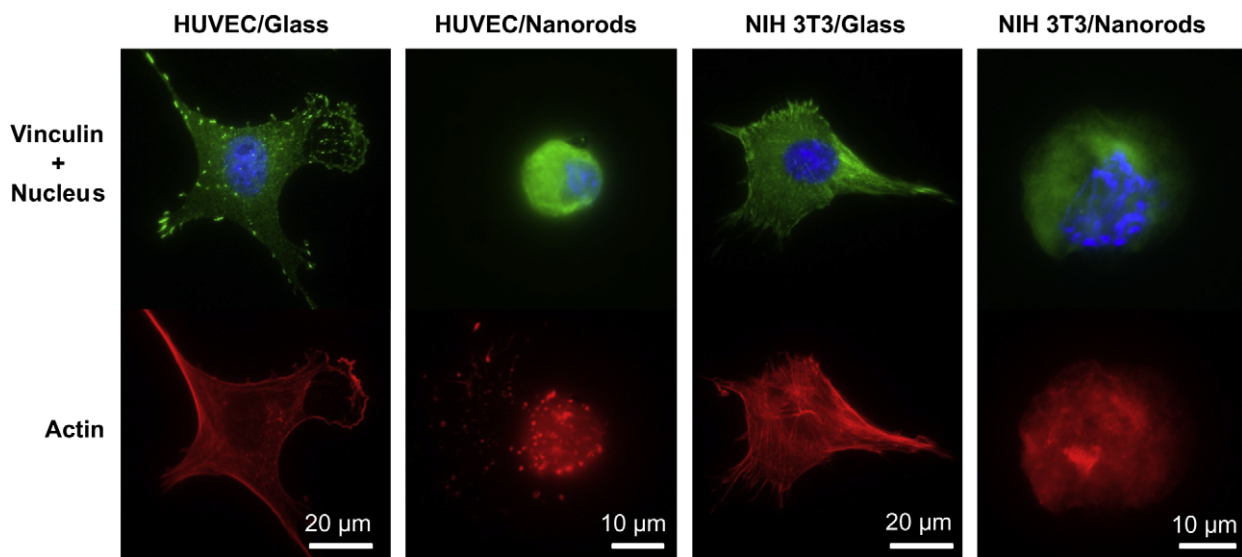


Fig. 2. Fluorescent microscopic images of HUVEC and NIH 3T3 on glass and nanorods. HUVEC and NIH 3T3 on glass assemble focal adhesions stained with vinculin (green) and actin stress fibers (red). Nuclei were stained with DAPI (blue). HUVEC and NIH 3T3 on nanorods are unable to spread and assemble focal adhesions and stress fibers.

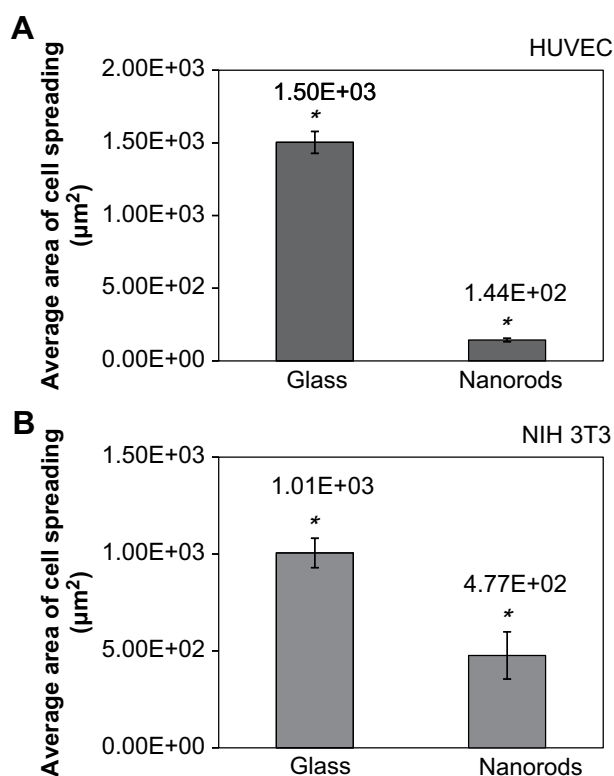


Fig. 3. The average area of cell spreading on glass and nanorods. (A) HUVEC on glass and nanorods ($n > 170$). (B) NIH 3T3 on glass and nanorods ($n > 110$). * indicates $p < 0.005$. Spreading area is significantly decreased on nanorods compared to glass. Bar indicates standard error of the mean (SEM). The data were pooled from three independent experiments.

2.2. Contact angle measurements

The contact angle of de-ionized water with surfaces was measured with a Ramé-Hart Goniometer and Ramé-Hart DROPImage Advanced Software using the sessile drop technique.

2.3. Cell culture

For control substrate 22 mm square glass cover slips were used (Corning, Inc., Lowell, MA). Before use, each substrate was sterilized with UV for 5 min and cleaned in 70% ethanol and de-ionized water. After drying substrates in air at room temperature, they were treated with 5 μg/ml human fibronectin (FN) (BD biosciences, Bedford, MA). After overnight incubation with FN at 4 °C, the substrates were washed twice with PBS. NIH 3T3 fibroblasts were cultured in DMEM (Mediatech, Inc., Herndon, VA) supplemented with 10% donor bovine serum (DBS) (Hyclone, Logan, UT). Human umbilical cord vein endothelial cells (HUVECs) were cultured in EBM-2 Basal Medium and EGM-2 Single Quot Kit (Lonza, Walkersville, MD). Cell suspensions of the same concentration and volume (i.e. same number of cells) were then seeded on each substrate.

2.4. Immunostaining and cell viability assay

After 24 h of cell seeding, non-adherent cells were removed with two gentle washes with PBS. The samples were fixed with 4% paraformaldehyde for 20 min and

washed several times with PBS. Fixed cells were immuno-stained for vinculin and stained for actin and nucleus using our previously reported methods [13]. Briefly, cells were fixed with 4% paraformaldehyde, permeabilized with 0.2% Triton X-100, and treated with mouse monoclonal anti-vinculin antibody (Sigma Aldrich, St. Louis, MO), followed by goat anti-mouse secondary antibody conjugated with Alexa Fluor 488 (Invitrogen, Eugene, OR). Actin was stained with phalloidin conjugated with Alexa Fluor 594 (Invitrogen, Eugene, OR) and nucleus was stained with 4'-6-diamidino-2-phenylindole (DAPI) (Sigma Aldrich, St. Louis, MO). Cells were then imaged on a Nikon TE 2000 epifluorescence microscope using GFP, Texas Red and DAPI filters. All images were collected using the NIS-Elements program (Nikon).

The live/dead viability/cytotoxicity kit for mammalian cells (Invitrogen, Eugene, OR) was used for quantifying adherent cell viability on each substrate. Cells were incubated at 30–45 min with calcein AM (2 μM for fibroblast and 4 μM for endothelial cells) and ethidium homodimer-1 (EthD-1) (4 μM for all types of the cells). Next, epifluorescence images of six to ten random fields were collected on a Nikon TE 2000 inverted microscope using a 10× lens for NIH 3T3 and HUVEC. The average number of cells adherent on each substrate, the number of adherent live cells (stained green with calcein AM) and adherent dead cells (stained red with EthD-1) were quantified from these images using the NIS-Elements program (Nikon). Three independent experiments of cell viability were performed and the data were pooled. The average area of cell spreading was determined from three independent experiments with statistical comparison using Student's *T*-test.

2.5. Time lapse imaging

Cells were pre-cultured on the patterned nanorods for 24 h as mentioned above. Before imaging, non-adherent cells were removed with two gentle washes with PBS and new media was added to the dish. Phase contrast imaging was performed for 6 h on the Nikon TE 2000 microscope with humidified incubator (In Vivo Scientific, St. Louis, MO). Images were collected every 5 min using a 10× objective.

2.6. Protein adsorption on nanorods and glass

Sterilized SiO₂ coated nanorod and glass substrates were prepared as outlined above. Both of the substrates were incubated with 10 μg/ml rhodamine fibronectin (Cytoskeleton, CO) diluted in PBS overnight and these dishes were washed with PBS several times. Five randomly taken 20× fluorescent images were collected with identical illumination and exposure time, and the fluorescent intensity was analyzed by the NIS-Element program (Nikon).

3. Results and discussion

3.1. Fabrication of SiO₂ coated nanorods

Many biomedical implants are made of temperature sensitive materials such as plastic. Hence, it is necessary to grow nanorods with techniques that do not require high temperature. Densely packed ZnO nanorods were fabricated with a low-temperature (95 °C) hydrothermal, solution-based growth method [36]. We next deposited nano-thin films of SiO₂ with controlled thickness, 50 Å, using PECVD at 50 °C according to our previously published methods [37]. Transmission electron microscopy (TEM) images of the resulting nanorods with 50 Å thickness of SiO₂ nano-films deposited are shown in Fig. 1A. The nanorods were randomly oriented in the upright direction, approximately 40–50 nm in diameter, 500 nm in height. The average spacing between nanorods was approximately 80–100 nm (Fig. 1A, white arrows). Importantly, the SiO₂ coatings were deposited uniformly on each nanorod free of any local defects, which was confirmed with TEM, local electrical conductance measurements, chemical wet-etching and photoluminescence

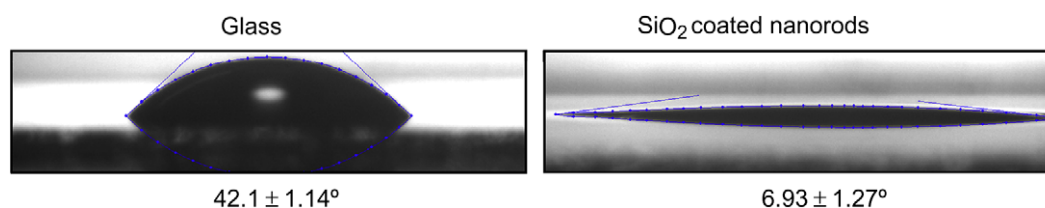


Fig. 4. Contact angles of water on glass and SiO₂ coated nanorods. The SiO₂ coated nanorod surface was hydrophilic, with average contact angle of 6.93±/–1.27° compared to glass of contact angle 42.1±/–1.14°. Error indicates the standard deviation (SD).

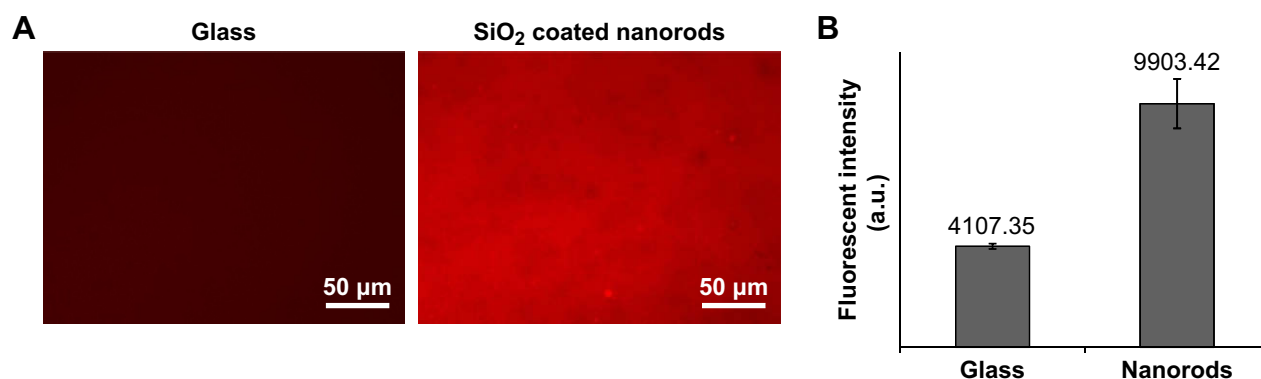


Fig. 5. Fluorescent images and intensity of rhodamine fibronectin coated glass and nanorods. (A) Representative fluorescent images of rhodamine fibronectin on glass and SiO₂ coated nanorods. (B) Plots show the average intensity profile pooled from five randomly taken images. Fibronectin adsorption is increased two-fold on nanorods compared to glass. Bar indicates SD.

intensity measurements [37]. Our technique thus resulted in randomly oriented, upright SiO₂ deposited nanorods that cover the surface with densely packed monolayers without any defects over cm length scales (Fig. 1C).

3.2. Decreased cell adhesion on SiO₂ coated nanorods

Cell adhesion and spreading occurs by the ligation of transmembrane integrins to ligands (such as fibronectin) immobilized on the surface. This is followed by clustering of the integrins at the nanoscale, and subsequent formation of multi-protein, micron-scale assemblies called focal adhesions [38]. Focal adhesions allow force transfer from the contractile acto-myosin cytoskeleton inside the cell to the outside surface, and this allows cells to adhere to and spread on the surface. If focal adhesions are not allowed to assemble in cells that depend on anchorage for survival, this leads to weak attachment to the surface, lack of cell spreading and subsequent apoptosis [39,40]. Therefore, the assembly of focal adhesions was next studied using immunofluorescence microscopy.

Human umbilical vein endothelial cells (HUVECs) and NIH 3T3 fibroblasts were cultured on SiO₂ nanorods which were pre-incubated with fibronectin overnight. Cells were fixed with paraformaldehyde and stained for vinculin, actin stress fibers and the nucleus. Both HUVECs and NIH 3T3 fibroblasts assembled vinculin-labeled focal adhesions on glass (Fig. 2). On the nanorod-coated surfaces, focal adhesions were not visible and cells were rounded and poorly spread (Fig. 2). Cells on nanorods were also unable to assemble contractile stress fibers. Consequently, the average area of cell spreading on nanorods was significantly decreased (Fig. 3) with a lack of focal adhesion and stress fiber formation. This result

suggests that cells are unable to spread and assemble focal adhesions on nanorods, which may cause apoptosis in these adhesion-dependent cells [39,40].

3.3. Protein adsorption on nanorods

Recent work by Spatz and co-workers showed that focal adhesion assembly requires the spacing between ligated integrins to be less than 70 nm [8,9]. A spacing of more than 73 nm between ligated integrins limits attachment, spreading, and actin stress fiber formation in fibroblasts. As the diameter of the SiO₂ nanorods is approximately 40–50 nm, local integrin clustering may occur but to a very limited extent given the vertical nature and small length (500 nm) of the nanorods. Due to the spacing of 80–100 nm, integrin clustering may not occur over multiple nanorods, preventing the assembly of contiguous focal adhesions on the micron length scale (Fig. 2).

Other possible explanations for the fact that cells cannot spread on nanorods are the super-hydrophobic nature of nanostructured surfaces such as ZnO nanorods [31,41]. Protein adsorption is decreased on super-hydrophobic surfaces which potentially can explain decreased adhesion. To address this question, we measured contact angles of SiO₂ coated nanorods. We found that SiO₂ coated nanorods were hydrophilic (Fig. 4: contact angle of 6.93±1.27° compared to glass of 42.1±1.14°). As fibronectin is known to adsorb successfully on hydrophilic surfaces [42], this result suggests that reduced matrix protein adsorption is likely not the reason for decreased adhesion.

To confirm this, we next measured the extent of fibronectin adsorption on nanorods (Fig. 5). Rhodamine-labeled fibronectin was deposited overnight on SiO₂ coated nanorods and flat glass

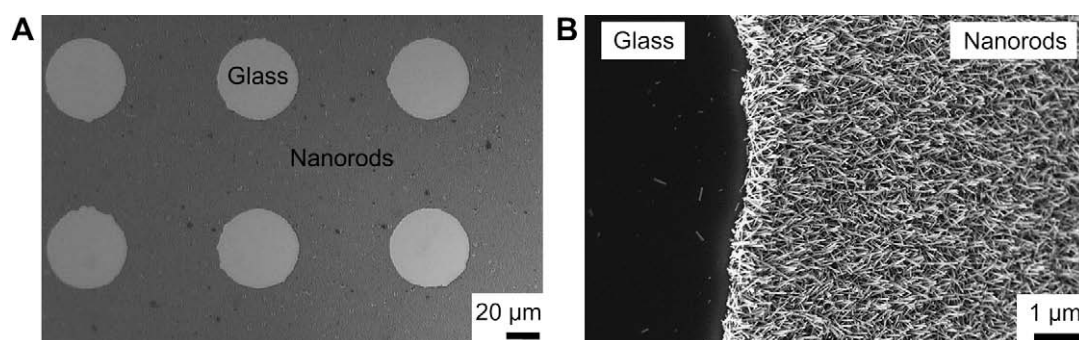


Fig. 6. SEM images of patterned nanorods. (A) Optical microscope image with 400× objective. (B) SEM image.

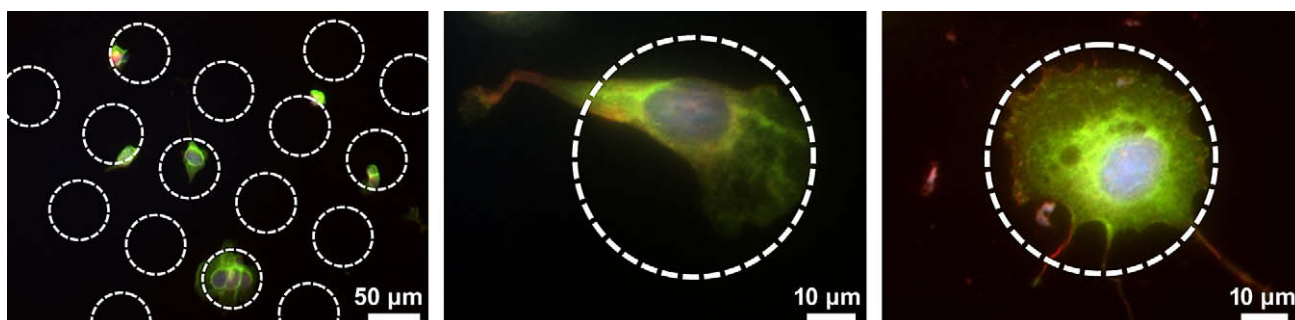


Fig. 7. NIH 3T3 fibroblasts on patterned SiO₂ coated nanorods. Fluorescent microscopic images showing that NIH 3T3 fibroblasts preferably attached on glass. Cells are stained for actin (red), vinculin (green) and nucleus (blue). Cells were confined on the flat circular regions. Dashed lines indicate the edge of patterns.

substrates. Fluorescent images of the rhodamine fibronectin adsorbed surface were captured and analyzed for differences in intensity. Interestingly, we found that fibronectin adsorption as measured by fluorescence intensity was increased two-fold on SiO₂ coated nanorods compared to glass. An increase in protein adsorption is to be expected given the increased surface area of the nanorods. The increase in fibronectin adsorption argues against large decreases in protein adsorption as being responsible for the observed reduction in cell adhesion. Importantly, fibronectin is known to adsorb in an active conformation on hydrophilic surfaces [42]. Given that all our experiments were carried out in 10% serum which allows the adsorption of other matrix proteins on the hydrophilic surface, and also promotes the secretion of fibronectin by the cells themselves, it is unlikely that decreased or abnormal

matrix protein adsorption plays a significant role in the observed response.

3.4. Spatial patterning of cell adhesion with nanorods

To investigate if it is feasible to pattern cell adhesion with nanorods, we spatially patterned nanorods using a low-temperature, and patterned growth method [36]. This method results in patterned nanorods that are not present inside circles, and are present outside in dense monolayers (Fig. 6). The diameter of circles was 50 μm and spacing between the circles was 40–60 μm. Nanorods were 50 nm in diameter and 500 nm in height. Fig. 7A and B shows that fibroblasts preferably adhered to the flat surface rather than to the nanorods after 48 h culture. Similar patterning was also observed with ZnO nanorods without SiO₂ coating (Fig. 1S in the supplementary information). Moreover, while the cells were confined to the circular regions on average, cells were frequently able to migrate from circle to circle by spanning the intervening nanorods (see Fig. 2S, and Movies 1S and 2S for an example of cell migration on patterned ZnO nanorods). This result suggests that spatially patterned nanorods provide a new way of dynamically patterning cells and therefore creating complex tissues.

3.5. Decreased cell survival on nanorods

The number of cells adherent on SiO₂ coated nanorods was significantly reduced (a reduction of 98% in fibroblasts, 82% in HUVECs) compared to cells on glass (Fig. 8A) after 24 h culture. Next, a live/dead viability/cytotoxicity kit for mammalian cells was used for quantifying adherent cell viability. The decrease in viability in cells on nanorods compared to that on glass was dramatic (Fig. 8B) with only one or two cells surviving on the SiO₂ nanorods for every 100 viable cells on glass. By culturing cells on glass in media that was incubated for 1 day, 3 days and 7 days with the nanorods, we confirmed that the cell death was not due to toxicity of unknown dissolving material from the nanorods (see Fig. 3S in the supplementary information). Therefore, these results suggest that densely packed nanorods have excellent anti-fouling potential by virtue of their topology.

4. Conclusion

Our results indicate that dense nanorod coatings are a powerful approach to eliminate cell adhesion and viability in anchorage-dependent cells, and a novel strategy for achieving anti-fouling. The mechanism is likely to be due to the lack of integrin clustering at the nanoscale.

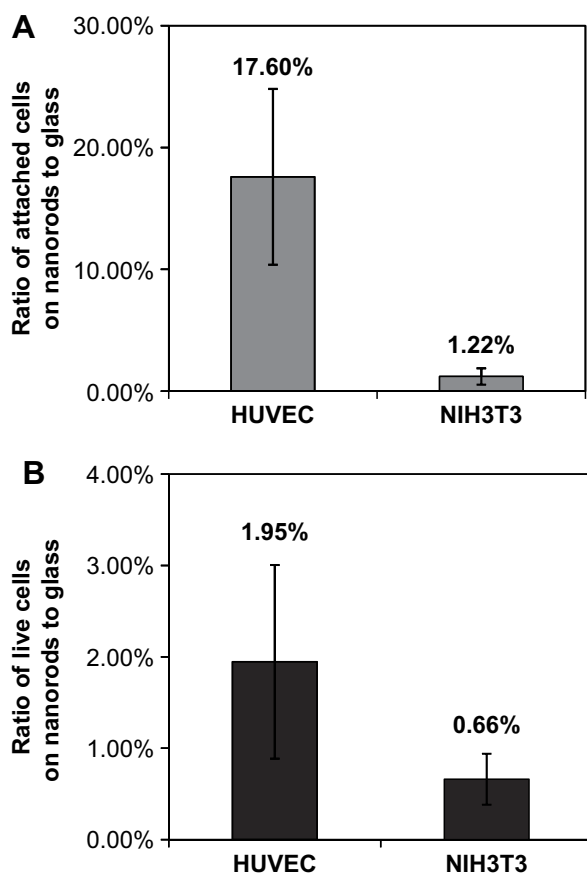


Fig. 8. Cell attachment and viability on nanorods. (A) The ratio of the number of attached cells on nanorods to that on glass. (B) The ratio of the number of live cells on nanorods to that on glass. ($n > 2500$ for HUVECs, $n > 1500$) for NIH 3T3. Cells are considerably reduced in numbers on nanorods. Bar indicates SEM.

Acknowledgements

We thank Dr. Chih-Yang Chang for analyzing SEM image and Lii-Cherng Leu for taking TEM image. This work was partially supported by a grant from the American Heart Association (0735203N) to TPL.

Appendix

Figures with essential colour discrimination. Certain figures in this article, in particular parts of Figs. 2, 5 and 7 are difficult to interpret in black and white. The full colour images can be found in the on-line version, at doi:10.1016/j.biomaterials.2009.05.028.

Appendix. Supplementary information

Supplementary material associated with this article can be found in the on-line version at doi:10.1016/j.biomaterials.2009.05.028.

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